Remarkable resilience of teeth (How are Teeth so Brittle yet so Resilient)

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**Recommended Citation**
How are Teeth so Brittle yet so Resilient?

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Tooth enamel is inherently weak, with fracture toughness comparable to glass, yet is remarkably resilient, surviving millions of functional contacts over a lifetime. We propose a novel microstructural mechanism of damage resistance, based on observations from ex situ loading of human and sea otter molars (teeth with strikingly similar structural features). Section views of the enamel implicate tufts, hypomineralized crack-like defects at the enamel–dentin junction, as primary fracture sources. We report a stabilization in the evolution of these defects, by 'stress shielding' from neighbors, as well as by inhibition of ensuing crack extension from prism interweaving (decussation) and by self healing. These factors, coupled with the capacity of the tooth configuration to limit the generation of tensile stresses in largely compressive biting, explain how teeth may absorb considerable damage over time without catastrophic failure—an outcome with strong implications concerning the adaptation of animal species to diet.
Mammalian dental enamel is highly brittle. Yet teeth can withstand high bite forces imposed thousands of times each day during chewing. In humans, these forces can reach up to 1000 N, and in great apes can be even higher (1,2). In contrast to natural crack-resistant laminate structures such as shells and nacre (3,4), enamel has a toughness close to that of glass (5,6), making it highly vulnerable to fracture (7,8). And fracture it does. Vertical cracks or 'lamellae' are commonly observed in erupted human teeth (9), progressing with age over millions of cycles. How do teeth sustain such cracks without catastrophic failure? Arguably the mammals in which crack-like defects matter most are those that break down hard food objects such as seeds (8,10-13) and mollusk shells (14,15). Such mammals include primates (incorporating humans and their hominin ancestors), bears, suoids (pigs and peccaries) and sea otters, each of which has converged on a low-crowned, blunt-cusped class of mammalian molar called 'bunodonty'. Perhaps the closest evolutionary parallel to hominins in the context of dental morphology are the sea otters (16), a comparison that we advance here to the microstructural level.

Recent studies are revealing how the enamel of bunodont teeth fractures under simulated biting, and are providing explicit relations for predicting critical forces for each fracture mode in terms of material properties and tooth/food contact geometry (8,17). Enamel thickness and cusp size emerge as key dimensions in these models. But what is the role of the underlying tooth microstructure? The literature on enamel morphology indicates a complex hierarchical structure consisting of bundles of hydroxyapatite nanocrystals within aligned prisms or rods weakly bound by thin protein sheaths (7,18). The prisms can undulate and cross each other (decussation), much like a basket weave
Enamel also contains arrays of intrinsic defects, principally hypocalcified fissures that emanate from the enamel–dentin junction (EDJ). These defects have the characteristics of 'closed cracks' filled with organic matter, and are known as 'tufts' for their wavy appearance within the prism microstructure (18,20,21). They are now believed to be a primary source of the fractures that ultimately develop in tooth enamel during extended function or overloading (22).

In this study we examine the fundamental nature of tooth resilience—and weakness—by conducting ex situ loading tests on extracted human and sea otter molars and examining the sources of fracture in sections of post-loaded specimens. We argue that enamel, while not macroscopically tough, nevertheless contains elements in its geometrical and microstructural complexion that confer a certain damage tolerance on the composite tooth structure. Such damage tolerance has long been argued to be an important factor in the adaptation of mammalian dentition to diet (13,23). It is also of interest in the context of biomimicry, in which dental researchers look to nature for inspiration in next-generation biomaterials design. We explore these connection further here.

**Experiment and Analysis**

Although biting forces are essentially compressive, weak tensile 'hoop' stresses form in the enamel coat and drive cracks around the tooth walls, eventually linking the cuspal surface and cervical margin in a 'longitudinal' fracture mode (22). Examples are shown in Fig. 1, after heavy overloading at the top surface with a flat disk indenter for (A) a human and (B) a sea otter molar. Two kinds of fractures are observed in Fig. 1:
‘radial–median’ cracks (R), initiating from the near-contact zone and propagating downward around the side walls; and ‘margin’ cracks (M), initiating from the cervical base and propagating upward. These same cracks find it difficult to enter the tougher, stress-shielded dentin (6,24); however, at higher loads they can link up to produce spalling of the enamel away from the dentin (25). The key observation is a substantial increase in load required to drive the cracks from inception to completion—hence the damage tolerance (22). This fracture resistance will be augmented by the filling of any slow growing fissures with organic fluids (9,26), thereby ‘gluing’ the crack walls together (self-healing) (27).

Experiments on slices sawn from extracted human molars reveal principal sources of enamel fracture. The slices were cut parallel to the tooth axis, immediately adjacent to a prominent cusp. The specimens were then loaded normally at the cusp with a flat disk. Figure 2 shows side views of the slice (A) before and (B) after loading. There is clear evidence of tuft extension parallel to the vertical loading direction in this specimen. In some specimens the tufts did not extend at all prior to failure, up to loads twice that attained in Fig. 2b, suggesting that there may be varying degrees of healing from the intrusive protein matter. Observe that several cracks have extended a considerable distance upward, more or less in unison, with one at the center ultimately breaking away from its neighbors upward toward the cuspal surface. This synchronized extension suggests a certain inbuilt stability and resilience to tooth fracture in normal dental function. Note also the disjointed nature of the extended tufts, indicative of crack inhibition at so-called 'Hunter–Schreger bands' delineating locations of periodic change in prism reorientation (18,19). Higher magnification views of this segmentation are
shown in Fig. 3, for cracks in both human and sea otter teeth.

To quantify the notion of synchronicity in tuft extension, we have conducted a plane strain finite element analysis for an enamel slice with periodic array of tufts under idealized cuspal loading, corresponding to the specimen geometry of Fig. 2. The tufts were considered as incipient vertical, equi-spaced cracks of length \( c \) and separation \( w \). The contact generates tensile stresses on the cracks, with maximum value \( \sigma \) at the EDJ and decreasing in magnitude along the load axis toward the top surface (28). These stresses fall off relatively slowly around the cusp walls, so adjacent tufts experience similar but slightly diminished fields. Crack driving forces, expressed as 'stress intensity factors' \( K \) evaluated from the crack-tip displacement fields (29,30) are plotted in Fig. 4 as a function of normalized crack lengths \( c/w \), for tufts 1, 2, 3 and 4, assuming all tufts to grow at the same value of \( c \) (solid curves, with label 1 referring to central tuft). These driving forces show a strong tendency to plateau out with crack extension, indicative of significant crack stabilization. Also plotted in Fig. 4 are comparative \( K(c/w) \) functions for a specimen containing just a single center tuft (i.e., no neighbors) within the same inhomogeneous contact stress field (heavy dashed curve) and within a hypothetical uniform tension stress field \( \sigma \) (light dashed curve). Note that the curve for the single tuft within the contact field still exhibits a plateau, indicating that some of the stabilization must come from the stress field inhomogeneity. The depression in plateau levels for the tufts in the array configuration must be attributable to stress shielding from neighbors. This enhanced stabilization accounts for the multiplicity of steady tuft extensions observed in Fig. 2b. While clearly an oversimplification of the complex loading geometries and microstructural complexions that characterize real tooth chewing function
in vivo, this simple model captures the essence of the crack stability.

The same stabilizing phenomenon is apparent in transverse sections of molars, i.e. sections cut normal to the tooth axis. Figure 5 shows images from (A) a human and (B) a sea otter molar after loading with a flat metal disk, at prescribed depths below the contact surface. The images include just a segment of the tooth periphery, with exposed dentin at top. Again, there is indication of synchronized tuft extension into the enamel from the EDJ. Some of these tufts (arrows) extend through the thickness of the enamel to the outer surface. Correlation with post-contact views of the side surface prior to sectioning reveal the crack outcrops in Fig. 5A to coincide with traces of margin cracks and in Fig. 5B with radial cracks, again implicating tufts as sources of failure.

A simple experiment was conducted to confirm the crack-like nature of the tufts. Figure 6 shows scanning electron micrographs of Vickers indentations placed adjacent to (A) a tuft and (B) a margin crack in a transverse section of a human molar. After applying the contact load, but before sectioning, this specimen was aged in water for one week and then allowed to dry. Figure 6A shows a corner crack \( CC' \) from one such Vickers indentation intersecting a tuft interface \( TT' \) (31). The tuft appears as a closed interface (i.e. filled with matter, presumably largely organic (9,32)), whereas the corner crack appears as an open interface (apart from some dislodged enamel fragments). In this example the intersected tuft has delaminated from the lower wall of the interface, confirming it as a weak path within the structure (33). In other cases (notably for more remote indentations) the tuft simply arrested the impinging corner crack without delamination, and in still others the corner crack penetrated the interface and extended into the adjacent enamel. These observations indicate that the filler material provides
some degree of protective interfacial bonding, with resultant inhibition of tuft extension. Figure 6B shows an analogous Vickers corner crack CC, intersecting an aged margin crack MM', in the same transverse section as in Fig. 6A. In this case the crack has penetrated the interface, indicating that organic material has intruded and solidified during aging, much as in the case of tufts. Corner cracks placed adjacent to newly formed margin cracks without aging in water always arrested at the interface.

**Discussion**

The present evidence suggests that the strategy of tooth survival is one of damage containment rather than avoidance. Tufts at the EDJ, while agents of weakness, are structured in arrays that provide internal shielding from applied stresses. Thus it is easy to initiate fractures within the enamel, but hard to drive these fractures to failure. Crack extension may be expected to progress steadily during a lifetime, which would explain why the dentition of older animals, including humans, tends to contain a population of visible crack-like defects (26). Since the process is gradual, the cracks will be continually replenished with protein-rich fluids, thereby slowing propagation even further by a mechanism of self healing similar to that observed in glass and other brittle materials (27). It is perhaps curious that although tufts have long been documented in the literature on dental histology (18), their potential role as sources of progressive damage has been totally overlooked. Decussation appears to offer some resistance to crack extension from the tufts, by a mechanism of stepwise crack arrest and penetration/reinitiation across the Hunter–Schreger bands, accounting for the waviness of the extended tufts. The toughness of enamel is over three times greater than that of constituent hydroxyapatite
crystallites within the prisms (34). The fracture patterns in Fig. 3 are reminiscent of those in brittle cross-ply laminate composites (35). Decussation occurs in different degrees in different animals, but generally with higher densities in the regions of the EDJ where the tufts originate (8). This implied resistance would appear to compensate somewhat for the intrinsic weakness of the inter-prism paths—without it, the toughness on pathways parallel to the prisms would be even lower than that of glass. The toughness of human enamel for crack paths perpendicular to the prisms is considerably higher relative to paths parallel to the prisms (5,6), making enamel more resistant to potentially deleterious lateral chipping (36).

The relationship of tooth form to diet remains a crucial issue in mammalian evolution. Such an understanding is vital to paleontologists and paleoanthropologists because teeth constitute a large portion of the fossil record (13). Arguments have been made asserting that such critical elements as enamel thickness, and to some extent tooth size, act as optimal organizations to frustrate tooth failure in relation to diet (8,13,17,22,37-39). However, it would seem that enamel microstructure, and especially tufts, are also important factors. The presence of tufts would appear to be a common feature of animal dentition, especially in primates: we have observed similar structures in enamel sections from chimps, orangutans and gorillas. Apart from their role in stress shielding, tuft arrays may protect the dentin by increasing the local enamel compliance, effectively 'grading' the material properties in the region of the EDJ (40). It is thus likely that the damage tolerance conferred by these defects is intricately tied to tooth morphology and function along the evolutionary path. The specific nature of the cuspal loading is another factor in determining which ensuing damage mode will dominate, and
this in turn depends on diet. For chewing on small hard particulates, the principal
damage mode is wear in the cuspal zone (8). Since wear is a local process relating to
asperity contact at the external surface, tuft configurations and decussation within the
enamel interior would be unlikely to confer much benefit. For feeding on large objects,
such as nuts and seeds, the greater threat is from longitudinal fracture (8,17). It is
noteworthy that mammals that consume large hard foods, including some primates, sea
otters and giant pandas, tend to show significant decussation in the inner sections of their
tooth enamel (41). In these cases, stress shielding, internal fissure self-healing, and prism
decussation adjacent to the EDJ—in conjunction with thick enamel and large cuspal
radius—offer inbuilt protection against catastrophic failure from dietary stresses.

Scientists in various disciplines may draw from the findings reported here. For
the large body of biologists concerned with the evolution of teeth, it suggests that largely
overlooked intrinsic structures like tufts may play a vital role in the interplay between
internal weakness of enamel and damage tolerance of the integrated tooth structure. Do
all enamels possess these features or are they linked predominantly to bunodonty? For
those biomaterials researchers who look to biomimicry for inspiration in the design of
synthetic enamel replacements (as well as of bone and soft tissue), the replication of
hierarchical fibrous microstructures incorporating weak internal interfaces presents a
fascinating challenge. Bio-inspired laminate structures that make use of weak internal
interfaces to confer toughness, analogous to shells and nacre, constitute just one example
of this design philosophy (42). To make any use of such biomimicry, it is crucial that we
first understand the underlying micromechanics that determine the structural resilience.
Unlike most artificial structures, mammalian systems have a natural inbuilt capacity for
self-healing—in the present case from protein-rich fluids in the dentition—that will tend to mitigate susceptibility to cyclic fatigue. Even this last element lies within the realms of possibility, as has been demonstrated by incorporation of microcapsulated crack-release healing agents into polymer microstructures (43). The future development of such damage-tolerant structures rests with the next generation of innovators.

**Materials and Methods**

Human molar and premolar teeth for testing were provided by Gary Schumacher and Sabine Dickens of the American Dental Association (ADA) laboratories at the National Institute of Standards and Technology (NIST). The teeth were extracted from male and female patients 18 to 25 years old and stored in aqueous solution. Intact specimens were selected and cleaned by Anthony Guiseppetti at the ADA. Cursory examination of the surfaces of the as-received teeth revealed the pre-existence of crack-like defects of various lengths extending from the cervical–enamel margins longitudinally toward the cuspal area. The average tooth width was 10 mm and crown height 7.5 mm. Approval to test these specimens was granted by the NIST Internal Review Board. The specimens were kept in aqueous solution after receipt and were kept moist during subsequent preparation and testing. Sea otter molar and premolar teeth were supplied by Jim Estes and Nate Dominy of the University of California at Santa Cruz (UCSC). These were from the skulls of deceased animals that had been dissected, cleaned and frozen by Andy Cunningham at UCSC before shipping to us in a dry state. Permission to transport and test the sea otter teeth was granted by Melissa Miller of the California Department of Fish and Game. Upon receipt, the teeth were extracted from the mandibles and stored in water. All remains of these animals are ultimately to be lodged at the California Academy of Sciences.

The photos of human and otter molars in Fig. 1 were obtained from specimens
after subjection to a compression fracture test (17). Individual teeth were mounted with their roots embedded in epoxy blocks, cusps uppermost. The mounted specimens were placed onto the platform of a mechanical testing machine, and kept moist during testing by continually squirting drops from a water bottle. A flat tungsten carbide rod was used to apply a vertical load to the most prominent cusp of each tooth. A video camera was used to view the progress of the cracks \textit{in situ} around the buccal or lingual walls immediately adjacent to the indented cusp during the loading. Oblique lighting was adjusted to provide optimum contrast at the fracture sites.

Sections through the teeth were made by conventional grinding and polishing. Images of the sections were obtained using reflection optical microscopy and scanning electron microscopy. The longitudinal slices in Figs. 2 and 3 were obtained with a diamond saw. Two cuts 1.5 mm apart were made either side of a selected cusp, parallel to the prospective vertical load axis. The resulting faces were polished to 1 µm finish with diamond paste. The slices were then mounted into epoxy blocks, and loaded with a tungsten carbide rod, as described above. Again, a video camera was used to photograph the tuft responses in the region of the EDJ. The transverse tooth sections in Figs. 5 and 6 were prepared by serial grinding and polishing of indented specimens, once more to 1 µm finish. These sections were polished to prescribed depths below the indented cusp.

Finite element analysis (FEA) was used to determine stress intensity factors for the crack array shown in the inset of Fig. 4. A mesh was set up for a curved enamel/dentin bilayer slice of depth 1.5 mm, with enamel thickness \(d = 1.8\) mm and outer radius 3 mm, using a standard ANSYS finite element package. Young's modulus and Poisson's ratio were taken as 90 GPa and 0.22 for enamel, 18 GPa and 0.35 for dentin. Some 9 cracks of size \(c\) and spacing \(w = 0.1\) mm incorporated into the enamel coat, labeled 1, 2, 3, 4, 5 as shown. A concentrated load was then applied at the top of the cusp by a rigid flat indenter. The mesh was scaled to smaller grid size in the vicinity of the crack tips, and the configuration refined until the solutions reached convergence. The
displacement fields near the tip of the cracks were thus determined, from which the stress intensity factor $K$, a measure of the generalized force driving the crack, was deconvoluted using standard fracture mechanics (44). Calculations of $K$ were made for each crack at specified values of $c$ (same for all cracks) with $w$ held fixed, and the curves in Fig. 4 thereby generated. Comparative calculations were made for a single center tuft in the absence of any neighbors, within the same contact stress field and within a hypothetical uniform stress, in order to separate contributions to crack stability from stress field inhomogeneity and near-neighbor shielding.

Acknowledgements

Funding from the GWU Research Enhancement Fund is gratefully acknowledged by PJC. We thank Sangwon Myoung for assistance with specimen preparation.
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Figure Legends

1. Fracture of teeth indented along a vertical axis with the flat end of a WC rod, (A) human molar, maximum load 390 N, (B) sea otter molar, maximum load 450 N. Radial–median (R) cracks have propagated part way downward from the contact zone, margin (M) cracks all the way upward from the cervical base. Some plastic flattening of the indented cusp is evident immediately beneath the indenter.

2. Optical micrographs showing how cracks grow from tufts at the EDJ in the vicinity of the vertical load axis, in human molar. Specimen is longitudinal slice of thickness 1.8 mm, (A) before and (B) after indentation with WC rod (upper cusp not shown). Cracks in (B) have extended upward from tufts toward the cuspal surface. The faintly visible fringes (Hunter–Schreger bands) mark changes in prism orientation.

3. Optical micrographs from longitudinal sections of (A) human and (B) sea otter molars, showing disruption of cracks at Hunter–Schreger bands. Location of field of view in (A) is near EDJ, immediately adjacent to the tooth axis; in (B) is some 2 mm below the cusp surface, midway between EDJ and outer enamel surface.

4. FEA calculations of stress intensity factor $K$ as function of reduced crack size $c/w$, for extension of periodic array of cracks in curved bilayer slice. Solid curves are calculations for crack arrays in inhomogeneous contact stress field, with $w/d = 0.1$ mm/1.8 mm = 0.055. Heavy dashed curve is for single crack in same stress field. Light dashed curve is for same single crack under uniform stress $\sigma$. 
5. Segment of transverse section view through molars: (A) human, loaded to 450 N and sectioned to depth 4.4 mm below the cuspal surface; (B) sea otter, loaded to 550 N and sectioned to depth 2.2 mm. Dentin is exposed at top of field of view. Cracks (arrows) appear to initiate from tufts.

6. Scanning electron micrographs from transverse section of human molar, showing intersection of Vickers corner crack CC' with (A) tuft TT', showing delamination at lower tuft interface along C'T and C'T', and (B) margin crack MM', showing penetration through interface to adjacent enamel.
Crack driving force, $K/\sigma w^{1/2}$

Crack size, $c/w$

Fig. 4
Fig. 5

A

Dentin

Enamel

B

Dentin

Enamel

Scale: 1 mm

Scale: 0.5 mm